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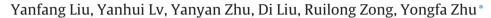
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Fluorine mediated photocatalytic activity of BiPO₄



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ABSTRACT

The fluorine doped BiPO₄ (F-BiPO₄) photocatalyst was synthesized via an in situ fluorination method. The lattice oxygen of BiPO₄ was substituted by fluorine, resulting in the increased polarizability and thus an increased induced dipole moment of BiPO₄, which can increase the separation efficiency of electron-hole pairs. The photocatalytic activity of BiPO₄ was enhanced by about 30% when the molar ratio of F/Bi was 0.03. Not only the discoloration rate of methylene blue (MB) but also the mineralization rates of MB and phenol were accelerated by F-doping. The enhancement of photocatalytic activity by F-doping could be attributed to the stronger adsorption ability of the substrate, the larger amount of active facet and higher separation efficiency of electron-hole pairs resulted from the increased polarizability. Overdoping would decrease the photocatalytic activity due to the increased recombination rate of electron-hole pairs at high defect concentration.

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1. Introduction

BiPO $_4$ is a newly discovered photocatalyst with more attractive UV light activity than TiO $_2$ (P25, Degussa) [1,2] and it has attracted increasing interest in the past few years [3–9]. It is not only efficient in decoloration of dye, but also superior in mineralization of the difficult degradable phenol. Synergy of BiPO $_4$ and adequate H $_2$ O $_2$ can raise the removal efficiency of phenol greatly [10]. Therefore, BiPO $_4$ is a promising photocatalyst and has much potential in remediation of environment. However, similar to TiO $_2$, BiPO $_4$ is also a wide band gap (ca. 3.85 eV) photocatalyst and the quantum efficiency is not high enough to meet the requirement of industrial application. Thus, it still needs to improve the photo-performances of BiPO $_4$ photocatalyst. How to increase its quantum efficiency and extend its light absorption range?

Doping is one of the methods to modify the bulk and surface characteristics and thus the light absorption range as well as the photocatalytic activity [11]. Both metal and nonmetal doping, as well as co-doping [12] have been investigated, and there are many excellent reviews related to doping [13,14]. Although cationic metal doping can induce a red shift in the band gap transition and may enhance the photocatalytic activity under both UV and visible light, some cationic metal dopants may also serve as recombination centers for photogenerated charge carriers and result in reduced photocatalytic activity [14]. Anionic nonmetal doping such as C [15], N [16,17], P [18], S [19,20], F [21–25], I [26] seems to

be more efficient and it is proved to have narrowed the band gap or improved the photocatalytic activity of different photocatalysts by inducing impurity states, which are supposed to be close to the valence band maximum. This can not only narrow the band gap, but also increase the separation efficiency of electronhole pairs and thus the photocatalytic activity. Among the anionic nonmetal dopants, fluorine is investigated most extensively and has been demonstrated to improve the photocatalytic activity of TiO_2 [25,27,28] and non TiO_2 -based photocatalysts [22,24,29–31]. Whether the photocatalytic activity can be improved and how much it can be improved by F-doping largely depend on the chemical nature of incorporated fluorine (such as adsorbed fluoride and lattice-doped fluorine) and the fluoride-mediated crystal modification and organization. Surface-fluorinated TiO₂ has been proved to be an effective surface modification method to improve its UV light photocatalytic activity [32], but it cannot induce strong visible light absorption [23]. Interestingly, Xing et al. [33] used a special structure of supported mesoporous catalyst to facilitate the surface fluorination of TiO2, and it was found that the surface fluorination enhanced the UV light photocatalytic activity and together with the Ti³⁺ generation led to the excellent adsorption capacity and visible light photocatalytic activity of TiO2. Lattice-doped fluorine can efficiently improve the photocatalytic activity [22,29] and sometimes can induce a red shift of the optical absorption edge of composite oxide photocatalyst [30]. For example, the fluorine interstitially doped ZnWO₄ can induce the distortion of WO₆ octahedron, resulting in an enhanced internal electric field and then an increased transfer rate of photogenerated electrons and holes to the catalyst surface, and thus an enhanced photocatalytic activity for degradation of Rhodamine B (RhB) and 4-chlorophenol (4-CP)

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[22,29]. Similar promotion in the photocatalytic activity over F-doped $\rm Bi_2WO_6$ was also reported [24]. Is it possible to improve the photocatalytic activity of the excellent UV light responsive BiPO₄, and further induce its visible light photocatalytic activity by F-doping? This is an inviting question and deserves intensive investigation.

The present work intends to explore the preparation and performance enhancement of F-BiPO₄. The effects of fluorine on the crystal structure and photocatalytic activity of BiPO₄ have been investigated. Besides, detailed mechanism for the improved photocatalytic activity by F-doping has been discussed.

2. Experimental

2.1. Preparation and characterization of F-BiPO₄ photocatalysts

BiPO $_4$ and F-BiPO $_4$ were synthesized via a hydrothermal process. All chemicals used were analytic grade reagents. In a typical procedure, 8 mmol of Bi(NO $_3$) $_3$ ·5H $_2$ O and 160 mL distilled water were put into a beaker under magnetic stirring. Then 28.8 mmol NaH $_2$ PO $_4$ ·2H $_2$ O and certain amount of NaF were added into the mixture and stirred for 1 h. The resulting suspension was transferred into a Teflon-lined stainless steel autoclave and maintained at 160 °C for 24 h. The products were washed three times with distilled water and dried at 120 °C for 8 h.

The products were characterized by X-ray diffraction (XRD) on Bruker D8-advance X-ray diffractometer at 40 kV and 40 mA for monochromatized Cu K α 1 (λ = 1.5418 Å) radiation. Morphologies of the prepared samples were further examined with transmission electron microscopy (TEM) by a Hitachi 7700 electron microscope operated at an accelerating voltage of 100 kV. UV-vis diffuse reflectance spectra (DRS) of the samples were measured by using Hitachi U-3010 UV-vis spectrophotometer. Raman spectra were recorded on a Horiba HR 800 spectrometer in the range of $100-1500\,\mathrm{cm}^{-1}$. The excitation light was $514\,\mathrm{nm}$. Chemical characterization of the sample surface was recorded with X-ray photoelectron spectroscopy (XPS ESCALAB 250Xi, Thermo Scientific). The charging effect was calibrated using the binding energy of C1s.

The photocurrent and Mott Schottky measurements were measured on an electrochemical system (CHI-660D, China). The asprepared samples for the fabrication of photoanode were obtained by mixing 1 mL of ethanol and 10 mg of as-prepared photocatalyst homogeneously. Then the homogeneous mixture was spread on the indium tin oxide (ITO) conducting glass. After the films were dried under ambient conditions, they were sintered in air at 200 °C for 5 h. The photocurrent and Mott Schottky measurements were measured with a standard three-electrode cell with a working electrode (as-prepared photocatalyst films on ITO conducting glass), a platinum wire as counter electrode, and a standard calomel electrode (SCE) as reference electrode. 0.1 M $\rm Na_2SO_4$ was used as the electrolyte solution. An 11 W ultraviolet germicidal lamp was used as the excitation light source.

2.2. Photocatalytic experiments

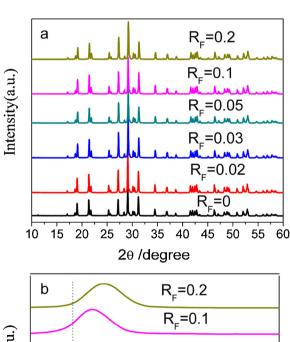
The photocatalytic activities were evaluated by the decomposition of MB under UV light (λ = 254 nm, 15 W) irradiation on an XPA photochemical reactor (Xujiang, Nanjing, China). A total of 25 mg of photocatalyst was dispersed in an aqueous solution of MB (50 mL, 0.02 mM). Before irradiation, the suspensions were first ultrasonicated for 10 min and then stirred for 30 min to ensure adsorption–desorption equilibrium between the catalysts and MB. At certain time intervals, 5 mL aliquots were sampled and centrifuged to remove the particles. The concentration of MB

was analyzed by recording the absorbance at the characteristic band of 663 nm using a Hitachi U-3010 UV–vis spectrophotometer. The intermediates formed in the degradation process of MB were identified by HPLC (Lumtech) system. Venusil XBP-C $_{18}$ (250 mm \times 4.6 mm i.d., 5 μ m) reversed phase column was used. The mobile phase consisted of two solutions namely A and B (60:40, v/v) with a flow rate of 1.0 mL min $^{-1}$. Solution A was made from 0.1 M ammonium acetate and acetic acid (pH 5.3), and solution B was acetonitrile. The UV detector was operated at 292 nm. Total organic carbon analyzer (Multi N/C 2100, Jena) was employed for mineralization degree analysis of MB and phenol solutions.

3. Results and discussion

3.1. Effect of F⁻ doping on the structure of BiPO₄

Fig. 1 showed the XRD patterns of BiPO₄ with different fluorine concentrations. The XRD profiles of the BiPO₄ and F-BiPO₄ samples could be indexed to the pure monoclinic phase (space group P21/n, JCPDS 15-0767. a=6.752 Å, b=6.933 Å, c=6.468 Å). The peaks at 27.15°, 29.10°, and 31.22° could be assigned to (200), (120) and (012) crystal planes of monoclinic BiPO₄, respectively. Although the crystal phase was not changed by fluorine doping, the relative peak intensities of the (200), (120) and (012) planes were changed. The ratio of 200/120 firstly increased and then decreased



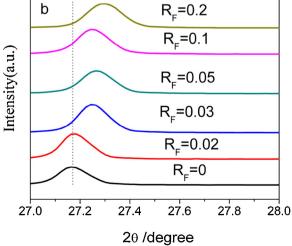


Fig. 1. (a) XRD patterns of BiPO₄ and F-BiPO₄; (b) diffraction peak positions of the $(2\,0\,0)$ plane in the range of 2θ = 27.0–28.0°.

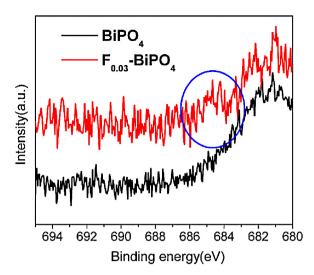


Fig. 2. The XPS spectra of BiPO₄ and F-BiPO₄ (F1s).

as $R_{\rm F}$ increased, and it reached the highest value when $R_{\rm F}$ was 0.03. So $F_{0.03}$ -BiPO₄ sample had the most obvious preferential crystallographic orientation along the [100] direction. This was probably caused by the selective adsorption of F^- on certain facet of BiPO₄. In addition to the preferential crystallographic orientation, a careful comparison of the (200) diffraction peak in the range of 27–28° (Fig. 1b) showed that the peak position of BiPO₄ with different fluorine contents shifted slightly toward a higher 2θ value. The same results were presented in other diffraction peaks. According to Bragg's law (Eq. (1)),

$$d_{(hkl)} = \frac{n\lambda}{2\sin\theta} \tag{1}$$

where $d_{(hkl)}$ is the distance between crystal planes of (hkl), λ is the X-ray wavelength, and θ is the diffraction angle of the crystal plane (hkl). The increase in 2θ value should result from the decrease in lattice parameters $(d(200) \, \text{value})$. The ionic radius of F^- (0.133 nm) is smaller than that of O^{2-} (0.140 nm), and the substitution of O^{2-} by F^- could result in the decrease of d(hkl) value and thus the increase in 2θ values. Therefore, the observed shift of the diffraction peak toward higher angle could be due to the substitution of lattice O by F. The existence of fluorine in F-BiPO $_4$ could also be confirmed by the XPS spectra (Fig. 2), although there was an overlap of F1s and Bi4p XPS spectra.

Fig. 3 showed Raman spectra of BiPO₄ with different fluorine concentrations. In all spectra, the observed intense band at $170\,\mathrm{cm^{-1}}$ with a shoulder at $231\,\mathrm{cm^{-1}}$ corresponds to the Bi–O stretching vibration. The bands centered at 1038 and $970\,\mathrm{cm^{-1}}$ are ascribed to the asymmetric (ν 3) and symmetric (ν 1) stretching vibrations of the PO₄ group, respectively. The ν 4 bending vibration modes of PO₄ groups occur at 554 and 596 cm⁻¹, and ν 2 bending vibration occur at 406 and 460 cm⁻¹ [9]. The positions of these Raman peaks were not changed but the intensities were increased by fluorine doping. This was also the case for fluorine doped Bi₂WO₆ [24]. The enhanced intensity may be caused by the increased polarizability and crystallinity.

3.2. Effect of F⁻ doping on the light absorption of BiPO₄

The DRS spectra of the BiPO₄ and F-BiPO₄ samples were recorded to characterize the optical properties (Fig. 4). Fluorine doping obviously affected light absorption characteristics of BiPO₄. The absorption spectra of the F-BiPO₄ samples showed a slight red shift in the band-gap transition. However, visible light absorption could not be realized by fluorine doping in the range of 0.02–0.2.

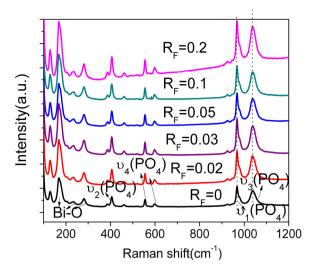


Fig. 3. Raman spectroscopy of BiPO₄ and F-BiPO₄.

The absorption above 400 nm confirmed that $F_{0,2}$ -BiPO₄ possessed the most defects among all the samples.

3.3. Enhancement of photocatalytic activity

The photoactivity of the BiPO₄ and F-BiPO₄ samples was evaluated by degradation of MB. Fig. 5 showed the relationship between photoactivity and the apparent rate reaction constant (k) of MB degradation over BiPO₄ and F-BiPO₄. Compared with pure BiPO₄, the photocatalytic activity of F-BiPO₄ firstly increased then decreased as the fluorine concentration increased from 0.02 to 0.2. The k reached the highest value and it was about 30% higher than the undoped sample when $R_{\rm F}$ was 0.03. However, further increasing the doped fluorine concentration could decrease the k value and k finally became smaller than that of the undoped sample when $R_{\rm F}$ was 0.2.

To investigate the degradation process of MB over BiPO₄ and F-BiPO₄ samples, the composition of MB intermediates were detected by HPLC (Fig. 6). Five peaks were detected from the spectra and they were assigned to MB, azure B(AB), azure A(AA), azure C(AC) and thionin (Th) [34,35]. From these intermediates, it could be concluded that MB was degraded through demethylation by both BiPO₄

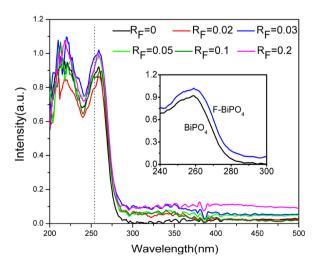


Fig. 4. DRS of BiPO₄ and F-BiPO₄ (the inset is the magnified figure for the BiPO₄ and $F_{0.03}$ -BiPO₄).

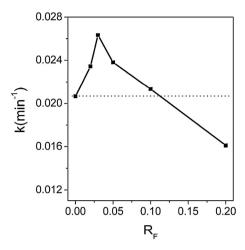


Fig. 5. Apparent rate constant (k) vs R_F .

and F-BiPO₄. Therefore, the degradation pathway of MB was not changed by fluorine doping.

TOC was conducted to elucidate the effects of fluorine on the mineralization ability of BiPO₄ photocatalysis. Fig. 7 showed that MB and phenol could be efficiently removed by BiPO₄ and F-BiPO₄. Therefore, F-BiPO₄ was more efficient than BiPO₄ in the mineralization of both colored dye and colorless pollutants.

3.4. Mechanism of enhanced photoactivity

There are several key factors influencing the photocatalytic capability of a semiconductor, including the absorption capacity of the incident photons, the photo excitation efficiency of electronhole pairs, the migration and separation rate of carriers, the redox capabilities of excited-state electrons and holes, as well as the nature of its surface/interface chemistry, which includes the surface energy and chemisorption properties of the photocatalyst [36].

Based on the DRS spectra of BiPO $_4$ and F-BiPO $_4$ samples, the absorption abilities of F $_{0.03}$ -BiPO $_4$ and F $_{0.2}$ -BiPO $_4$ above 254 nm (the irradiation light wavelength) were similar, whereas the photocatalytic activity of the two samples were the highest and lowest among all the samples, respectively. Besides, F $_{0.02}$ -BiPO $_4$ possessed the lowest absorption ability among the other four samples, whereas its photocatalytic activity was almost

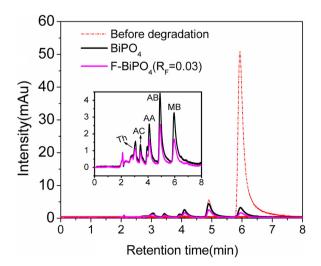


Fig. 6. HPLC spectrum of MB solution before and after photocatalysis (30 min of reaction time) over $BiPO_4$ and $F-BiPO_4$ ($R_F=0.03$) (The inset is the magnified figure for the after photocatalysis solutions).

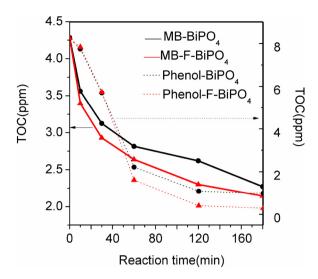


Fig. 7. Changes of TOC during the photocatalytic degradation of MB and phenol over $BiPO_4$ and F- $BiPO_4$ (R_F = 0.03).

the highest among them. Thus the absorption capacity of the incident photons and the photo excitation efficiency of electronhole pairs were not the decisive factors for their photocatalytic activity.

Photocurrent is a direct reflection of the mobility capability of electrons generated in the photocatalyst, and the rate should directly correlate with the photocatalytic activity of the material [37]. Fig. 8 showed the photocurrent of BiPO₄ and F-BiPO₄ samples generated in suspensions under UV light. It was noted that except for F_{0.2}-BiPO₄, the photocurrent generated from the fluorine doped BiPO₄ were all larger than that of pure BiPO₄. The increased photocurrent of F-BiPO₄ could be attributed to the higher separation efficiency of photogenerated electron-hole pairs. As for F_{0.2}-BiPO₄, its relatively low photocurrent might be caused by the fast recombination rate of photogenerated electron-hole pairs [38].

To understand the difference in electronic properties of $BiPO_4$ and $F-BiPO_4$ electrodes, Mott Schottky (MS) measurements (Fig. 9) were performed in the darkness by using the impedance technique. As we know from the classical Mott Schottky theory, when the doping semiconductor space-charge region is depleted as it is under

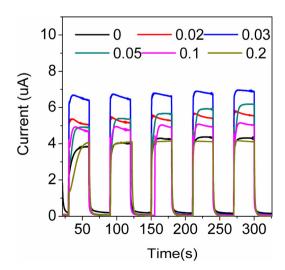


Fig. 8. Photocurrent response for $BiPO_4$ with different R_F .

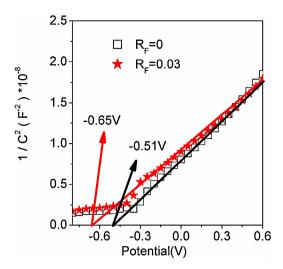


Fig. 9. Mott Schottky (MS) plots of the different catalysts film electrodes.

biasing conditions, the capacitance of space charge region can be described after simplification as Eq. (2)

$$\frac{1}{C^2} = \frac{2}{eN_{\rm d}\varepsilon_0\varepsilon} \left| V - V_{\rm fb} \right| + \frac{1}{C_{\rm H}^2} \tag{2}$$

where e is the electronic charge, ε_0 is the permittivity of free space, ε is the dielectric constant of the BiPO₄ material, $N_{\rm d}$ is the dopant (donor or acceptor) concentration, V is the applied potential, $V_{\rm fb}$ is the flat-band potential and $C_{\rm H}$ is the Helmholtz capacitance of the electrode [39,40]. The flat-band potential $V_{\rm fb}$ can be obtained from the x-intercept of the linear region of the curve $1/C^2$ vs V, and the dopant density $N_{\rm d}$ can be determined from the slope of the linear region. Based on Fig. 9, the flat-band potential of BiPO₄ and F-BiPO₄

 $(R_{\rm F}\,0.03)$ electrodes were $-0.51\,\rm V$ and $-0.65\,\rm V$ vs SCE, respectively. It is generally known that the conduction band minimum (CBM) in many n-type semiconductors is more negative by approximately 0.1 V than the flat band potential [41]. Therefore, F-BiPO₄ showed a negative shift of the conduction band compared with the pure BiPO₄. Based on the band gap (Supporting Information Fig. S1), the estimated positions of valence band maximum (VBM) were 3.29 and 3.03 V vs SCE for BiPO₄ and F-BiPO₄, respectively. The negative shift of CBM and VBM would be in relation to the increase of the reduction potential of the photogenerated electrons and the decrease of the oxidation potential of holes. However, the lowered oxidation potential of F-BiPO₄ was still a little bit higher than or equal to that of TiO₂ (2.9 V vs NHE [42], thus ca. 2.7 V vs SCE), which guaranteed its strong oxidation power for MB and phenol. In addition, the slope of the linear region for F-BiPO₄ showed a smaller value than that of BiPO₄, indicating higher donor density for the doped electrode. According to photocatalytic mechanism, the higher the donor density for the doped electrode, the faster the photocatalytic degradation rate [29]. This was further proved by the photocurrent. Therefore, the enhancement of photocatalytic activity could be attributed to the higher separation efficiency of electron-hole pairs, and a large number of holes participated in the photocatalytic process caused by fluorine.

In terms of the morphology and structure of the photocatalyst, several factors could influence the separation and recombination efficiency of photogenerated electron-hole pairs. These factors include particle sizes [8], crystallinity [36], dipole moment [43] and defects [38]. Small particle sizes are beneficial to decrease the electron hole migration distance and high crystallinity will reduce electron hole recombination rate. On the basis of XRD, the crystallinity of the samples was fine. Based on TEM (Fig. 10) images of BiPO₄ and F-BiPO₄, it could be seen that the samples were all nanorods. As for the 1D nanostructure, increasing the aspect ratio (the ratio of length to diameter or surface to volume) generally

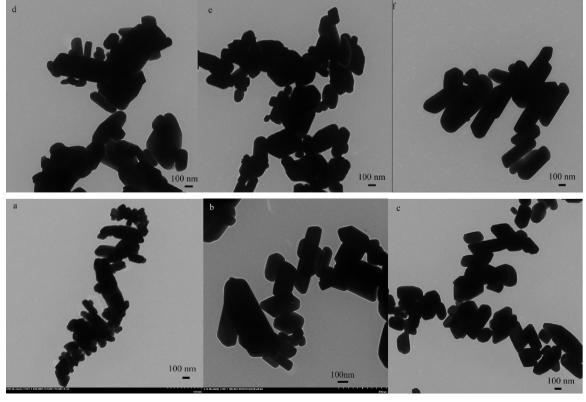


Fig. 10. TEM images of BiPO₄ and F-BiPO₄. $R_F = 0$ (a), 0.02 (b), 0.03 (c), 0.05 (d), 0.1 (e), 0.2 (f).

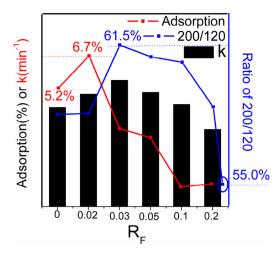


Fig. 11. Adsorption percentage of MB on F-BiPO₄ after equilibrium, ratio of 200/120 and the apparent rate constant (k) with different R_F (the point in circle is calculated from JCPDS 15-0767).

results in an enhancement of activity [36]. In our work, BiPO₄ samples with different fluorine concentrations possess similar morphology and aspect ratio, except that F_{0.2}-BiPO₄ had relatively larger aspect ratio. The larger aspect ratio did not result in higher photocatalytic activity. On the contrary, F_{0.2}-BiPO₄ had the lowest activity. Therefore, particle size was not the decisive factor for the photocatalytic activity. Raman spectra revealed that the polarizability could be increased by F-doping. The increased polarizability would increase the induced dipole moment and thus increase separation efficiency of photogenerated electron-hole pairs. Thus the increased polarizability was partly responsible for the increased photocatalytic activity. It could be concluded from the DRS that there were the most defects in F_{0.2}-BiPO₄. It is stated that defects form the trapping centers of photoinduced electrons and holes at low defect concentration and change to recombination centers at high defect concentration [38]. In our case, the high concentration defect induced by fluorine doping resulted in fast recombination rate of photogenerated electron-hole pairs and thus low photocatalytic activity. To conclude, the separation and recombination rates of photogenerated electron-hole pairs were mainly mediated by the induced dipole moment and defects by F-doping.

Apart from the above factors, the nature of photocatalyst's surface/interface chemistry is another dominating factor influencing the photocatalytic capability of a photocatalyst. In general, a stronger adsorption ability of the substrate [44,45] and larger percentage of highly reactive facets will yield higher photocatalytic activity [46-49]. As for the fluorine doped BiPO₄, adsorption played an important role on their photocatalytic activity. Based on Fig. 11, it was clear that the adsorption of MB on BiPO4 was altered by fluorine doping. Compared with BiPO₄, the amount of MB adsorbed on F-BiPO₄ firstly increased then deceased as the fluorine concentration increased from 0.02 to 0.2. The plateau was reached when $R_{\rm F}$ was 0.02. When $R_{\rm F}$ increased to 0.2, the adsorption content decreased to 1/5 of the pure BiPO₄. This result might seem strange at first glance and one might even doubt whether the adsorptiondesorption equilibrium had been achieved. To clarify this question, we examined the change of adsorption percentage over time in dark and found that the percentage of MB adsorbed on BiPO₄ at 30 min and 90 min were similar, and the relationship between adsorption percentage and R_F was the same. The "first increase then decease" phenomenon was speculated to be related with the state of fluorine in BiPO₄. When the amount of added fluoride was small, most of the fluoride was adsorbed on the surface of BiPO₄, making the photocatalyst more negative. MB molecule, as a cationic dye, would be much easier to be adsorbed on the negative surface. When the amount of added fluoride increased, most of the F- went deep into the lattice of BiPO₄, substituted the lattice O²and made the surface of BiPO₄ less negative. Thus, the cationic MB molecule would be less likely to adsorb on such surfaces. The adsorption ability of F-BiPO₄ had some correlation with the photocatalytic activity. In general, the stronger the adsorption ability was, the higher the photocatalytic activity became. However, there also existed some exceptional cases. The low adsorption ability of $F_{0.03}$ -BiPO₄ did not result in low photocatalytic activity. Besides, F_{0.1}-BiPO₄, possessing similar or even lower adsorption ability for MB than $F_{0,2}$ -BiPO₄, had higher photocatalytic activity than $F_{0,2}$ -BiPO₄. Furthermore, it could be seen that the maximum amount of MB adsorbed on F-BiPO₄ was less than 10%. This small value indicated that adsorption ability could not be the decisive factor for the high photocatalytic activity of F-BiPO₄. In general, a higher surface energy yields higher catalytic activity. So much effort has been made in synthesizing photocatalysts with highly reactive facets. Based on the XRD, it can be seen that the ratio of 200/120 had some correlation with the photocatalytic activity. When $R_{\rm F}$ was in the range of 0-0.05, the larger the ratio of 200/120 became, the higher the photocatalytic activity was. So there must be more active facets when BiPO₄ nanorods grow along the [100] direction. The larger amount of active facets would contribute to the enhanced photocatalytic activity of F-BiPO₄.

4. Conclusion

The photocatalytic activity of BiPO₄ could be improved by F-doping. This could be attributed to the stronger adsorption ability of the substrate, the larger amount of active facet and higher separation efficiency of electron-hole pairs. The photocatalytic process would be inhibited by high dopant concentration (\geq 0.2) due to the increased recombination rate of electron-hole pairs at high defect concentration. Although a red shift of the optical absorption edge of BiPO₄ was induced by F-doping, visible light absorption cannot be achieved.

Acknowledgments

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Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at http://dx.doi.org/10.1016/j.apcatb.2013.09.050.

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